

The Reciprocal Lattice

-- continued

A set of wave vectors \mathbf{K} that ensure

$$e^{i\mathbf{K} \cdot \mathbf{R}} = 1$$

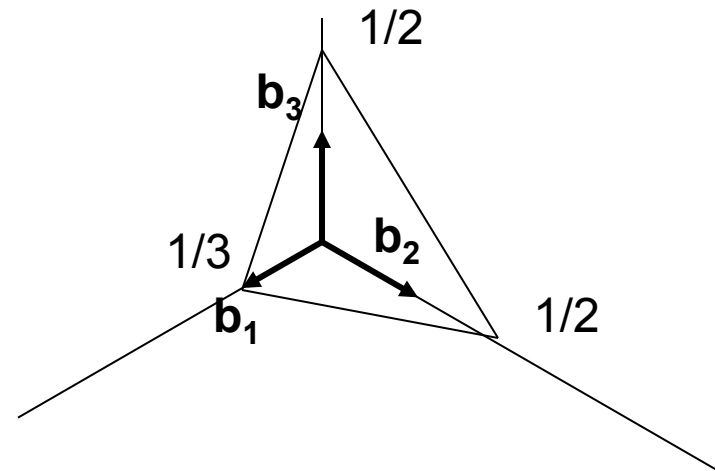
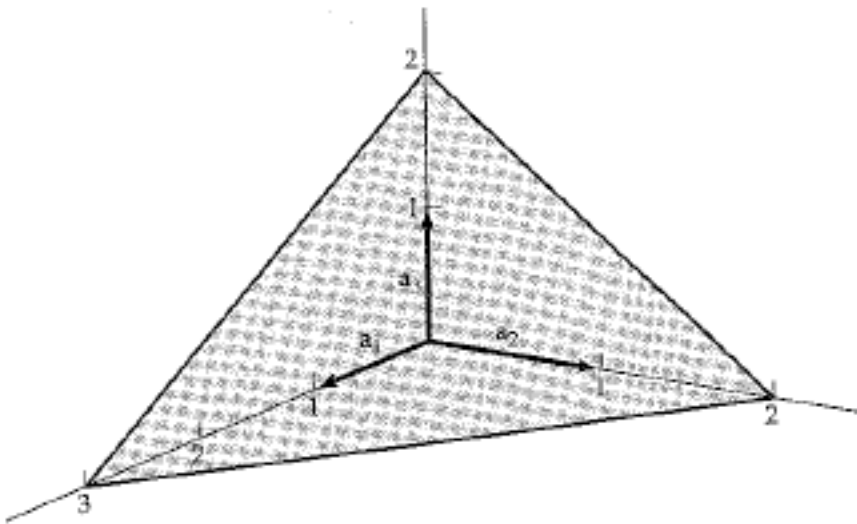
Direct lattice (Bravais lattice): $\mathbf{R} = n_1\mathbf{a}_1 + n_2\mathbf{a}_2 + n_3\mathbf{a}_3$

Reciprocal lattice: $\mathbf{K} = k_1\mathbf{b}_1 + k_2\mathbf{b}_2 + k_3\mathbf{b}_3$

Application:

Name Lattice Planes - Miller Indices

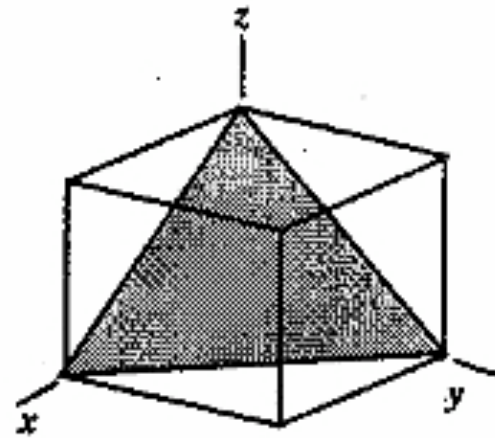
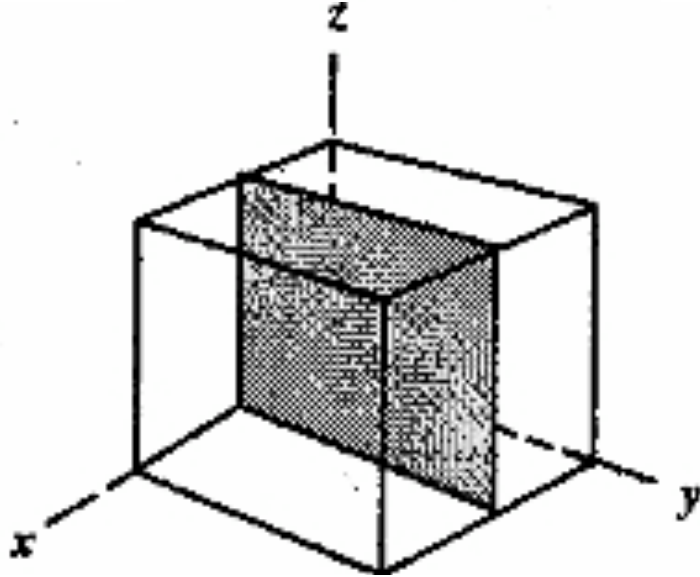
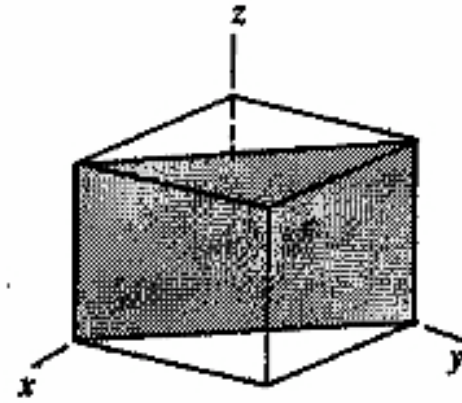
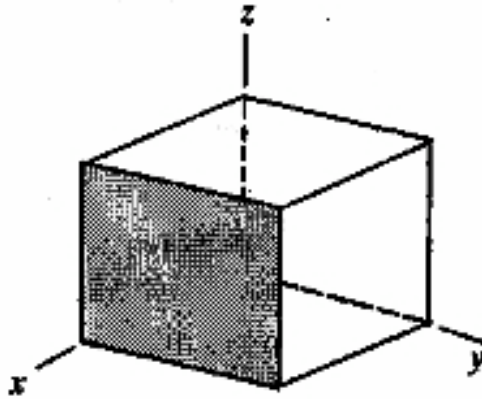
Direct lattice

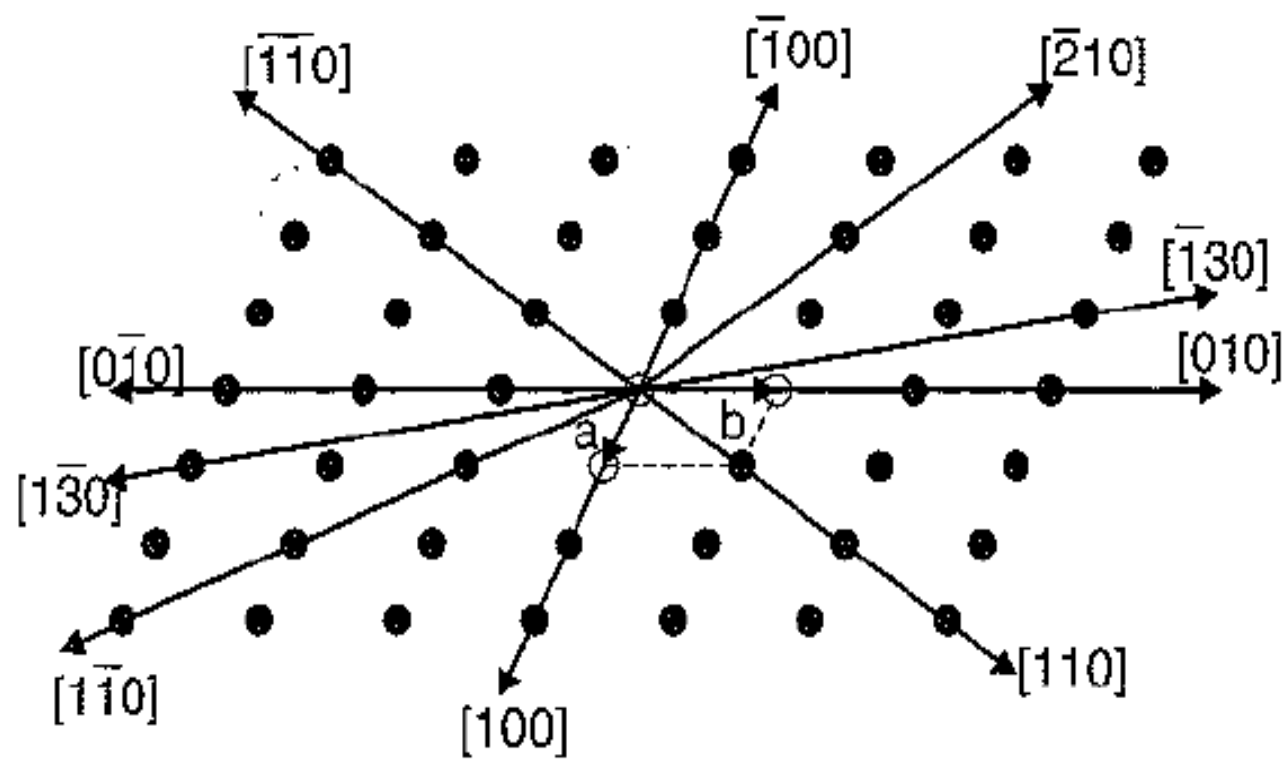


The smallest three integers
having the same ratio as
 $1/3 : 1/2 : 1/2 = 2 : 3 : 3$

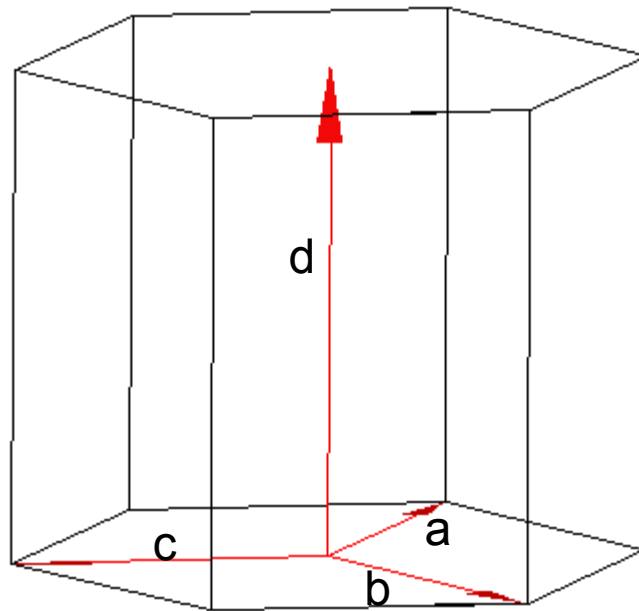
The plane is called (233) plane
- Miller Indices

Three Miller Indices (hkl)



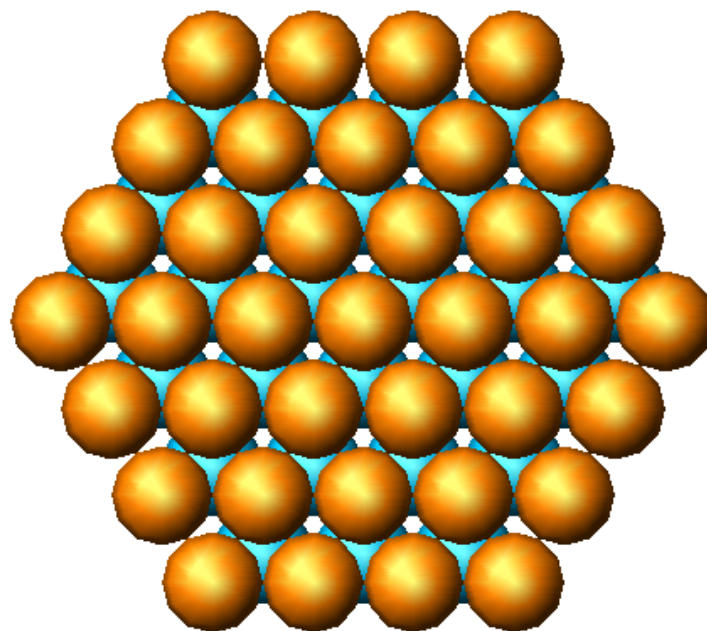
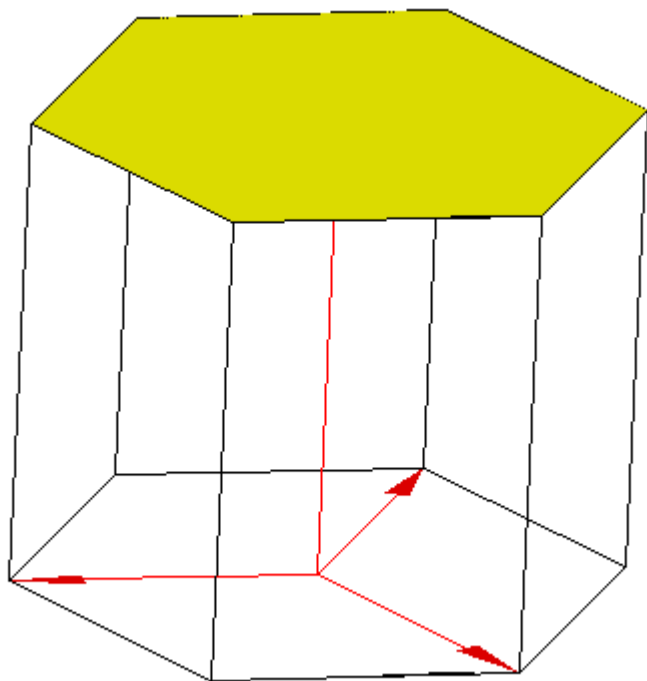


Four Miller-Bravais Indices (hkil)

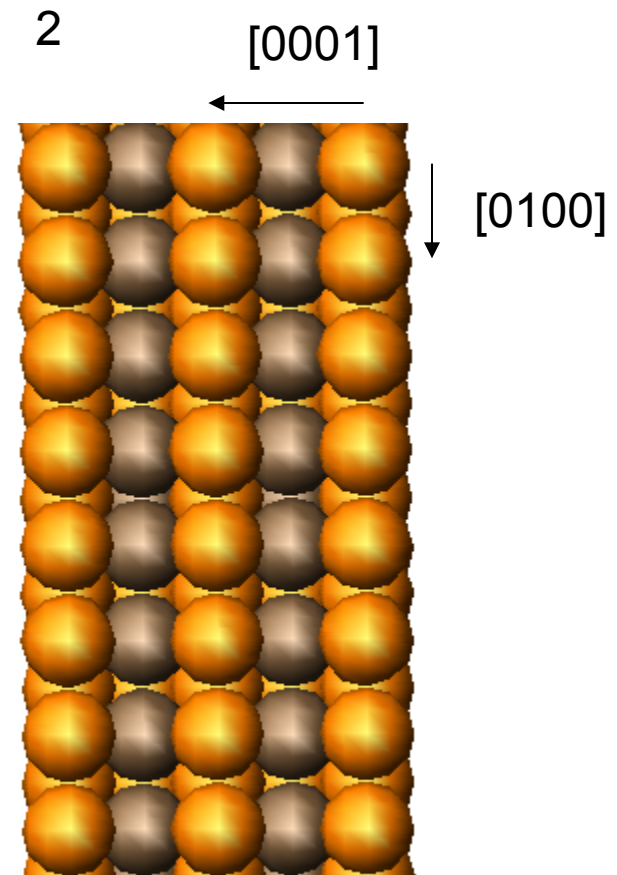
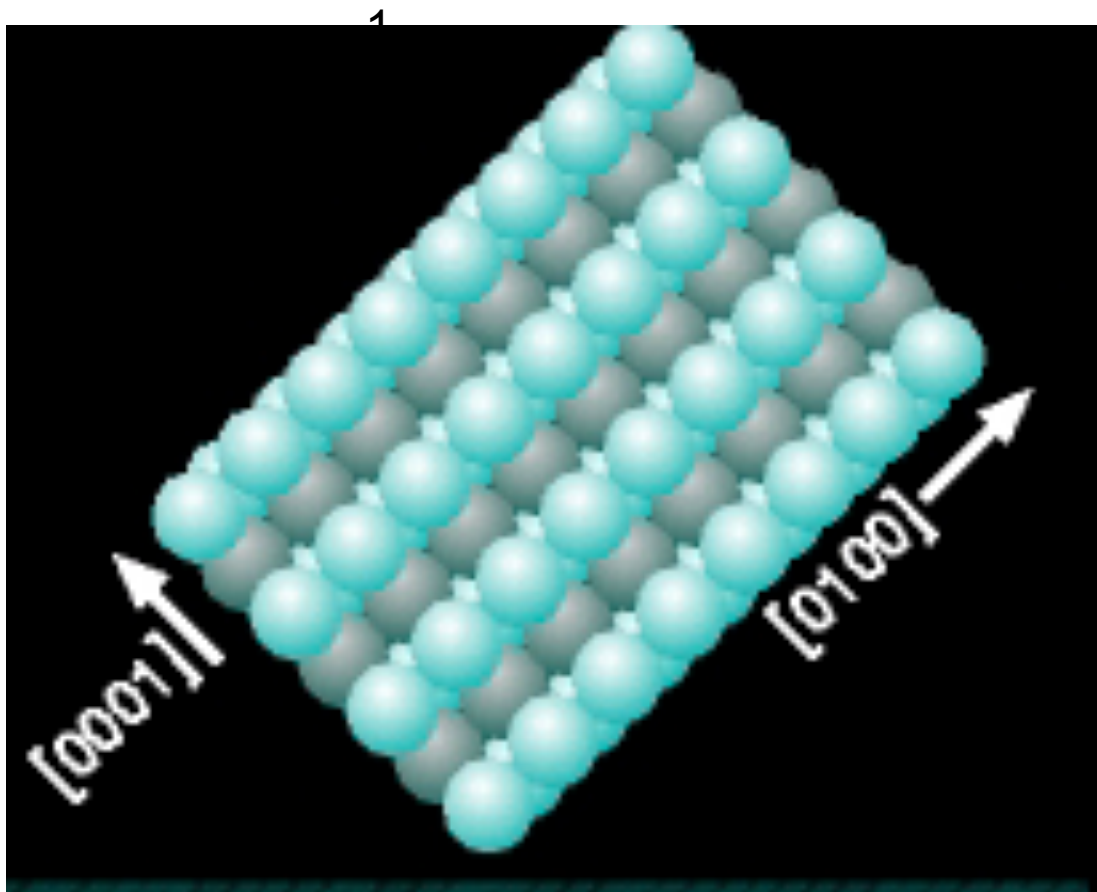
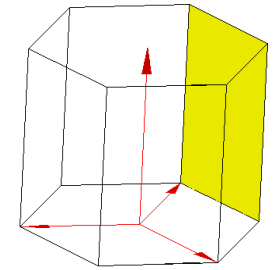


$$h + k + i = 0$$

(0001)

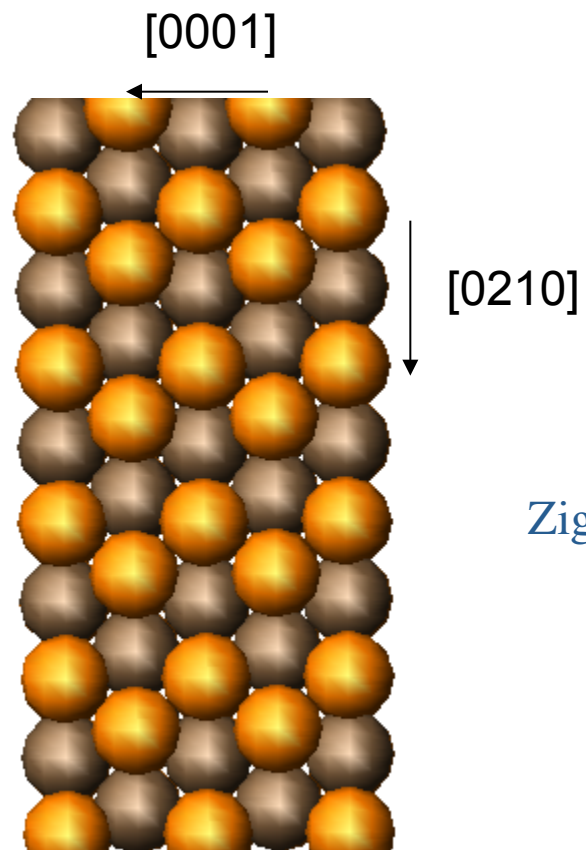


$(10\bar{1}0)$

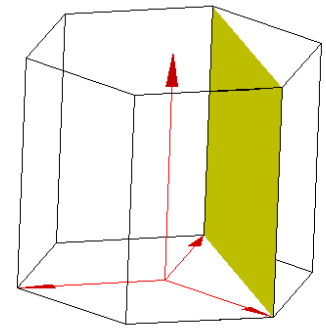
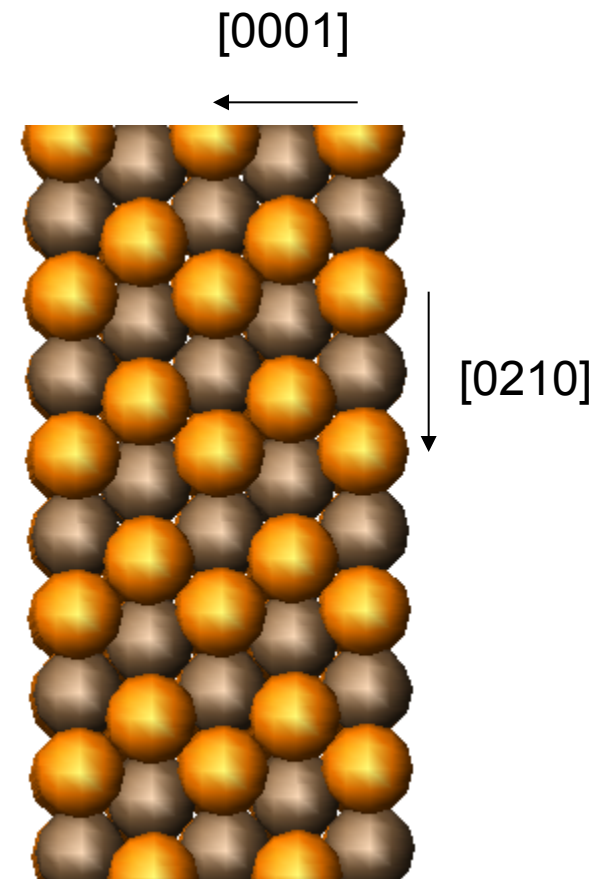


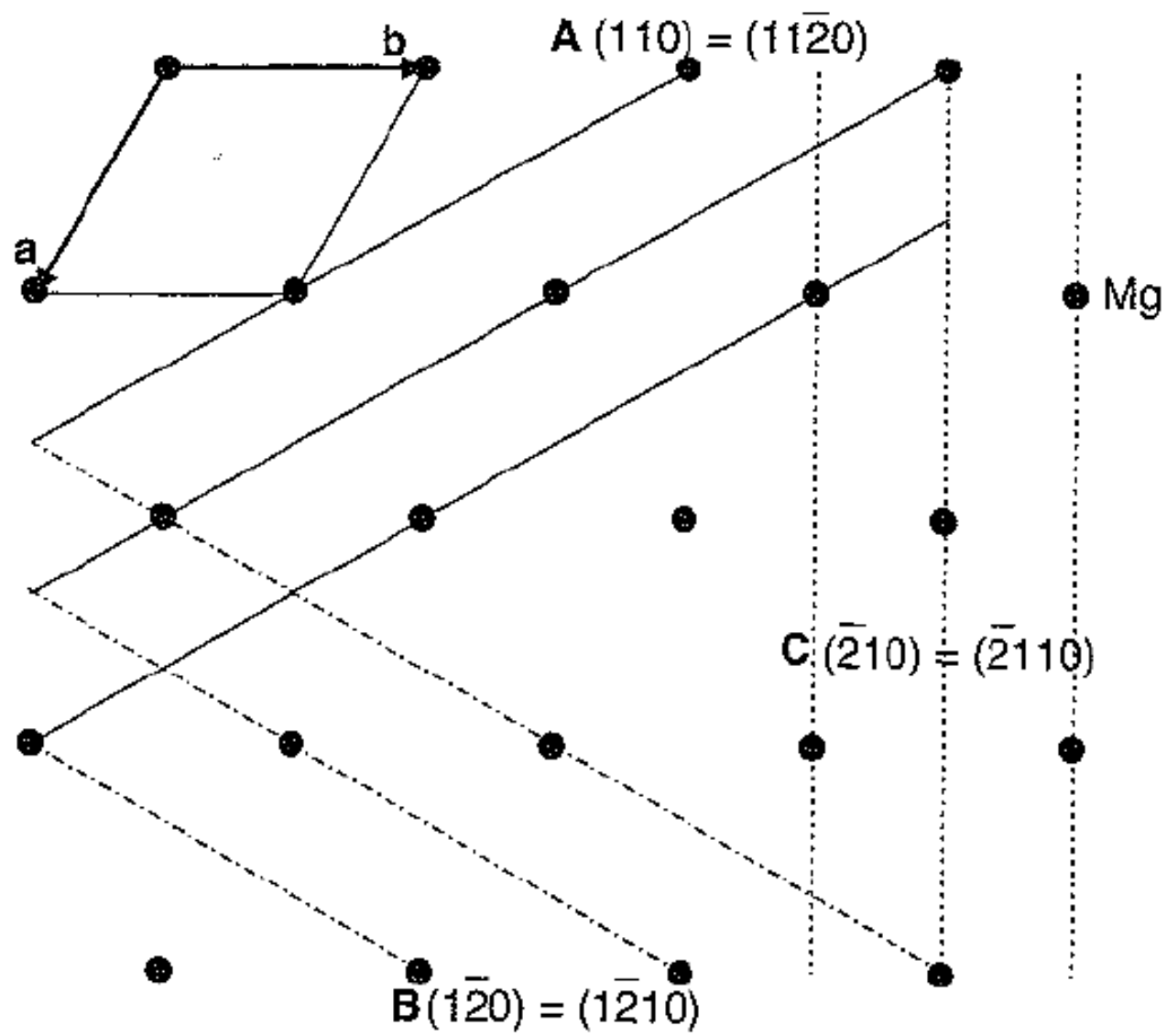
$(11\bar{2}0)$

1



2



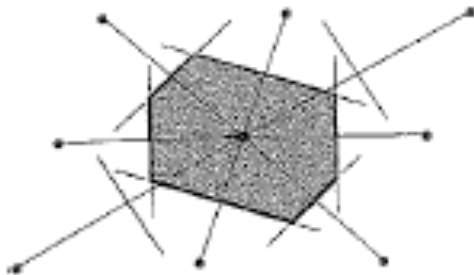


Application:

Construction of Brillouin Zones

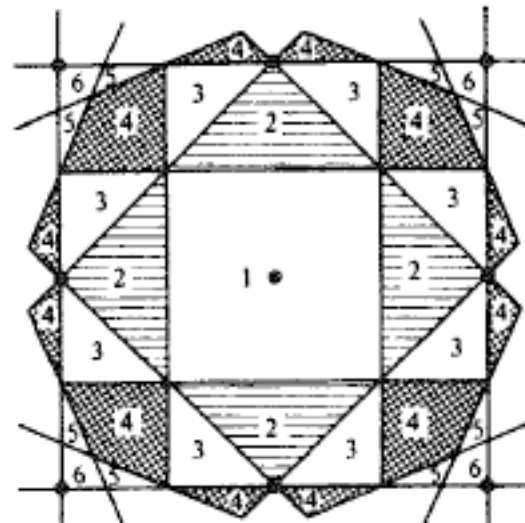
In Bravais lattices
(direct lattices)

Wigner-Seitz cell



In reciprocal lattices

Wigner-Seitz cell
= **first Brillouin Zone**



Determination of Crystal Structures

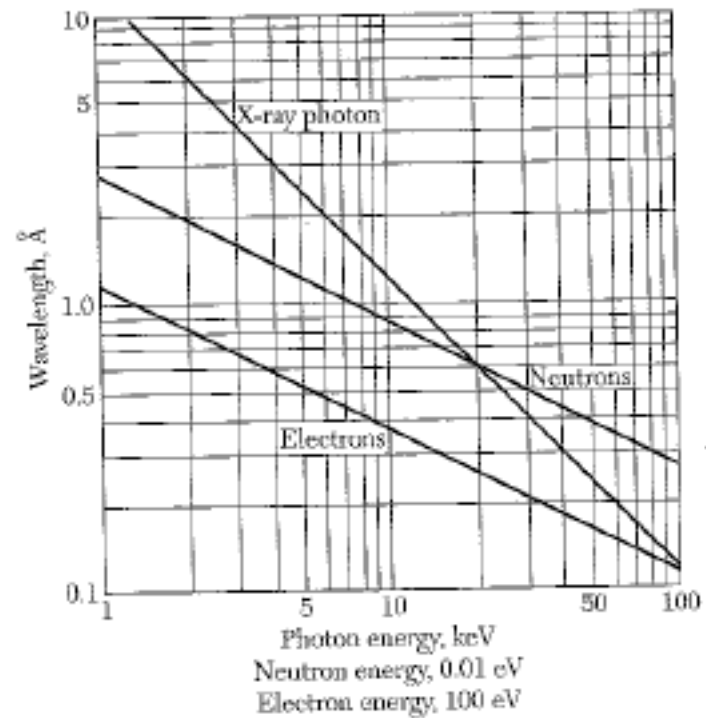
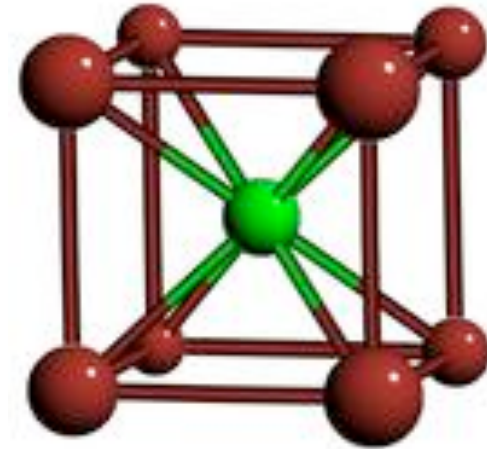
	2D	3D
Crystal structures	5	7

Distance between atoms
in the order of

$$\text{\AA} \sim 10^{-8} \text{ cm} \sim 10^{-10} \text{ m}$$

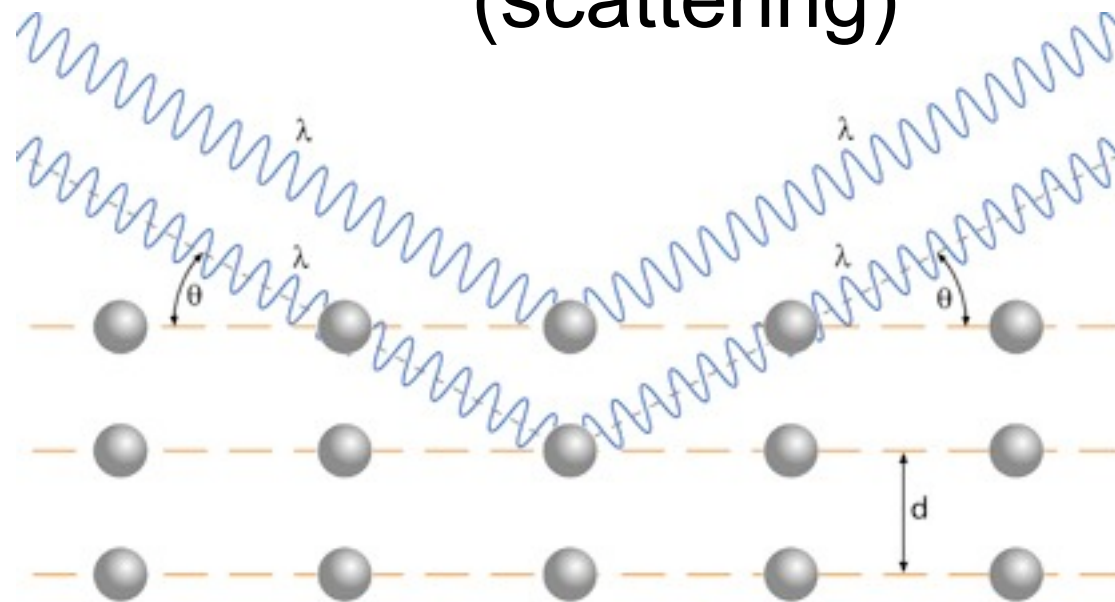
requires

Wave length $\lambda \sim \text{\AA}$



Determination of Crystal Structures:

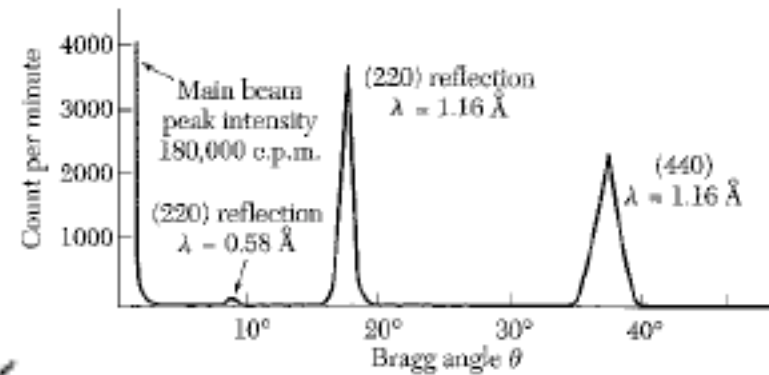
X-ray diffraction (scattering)



Bragg Law:

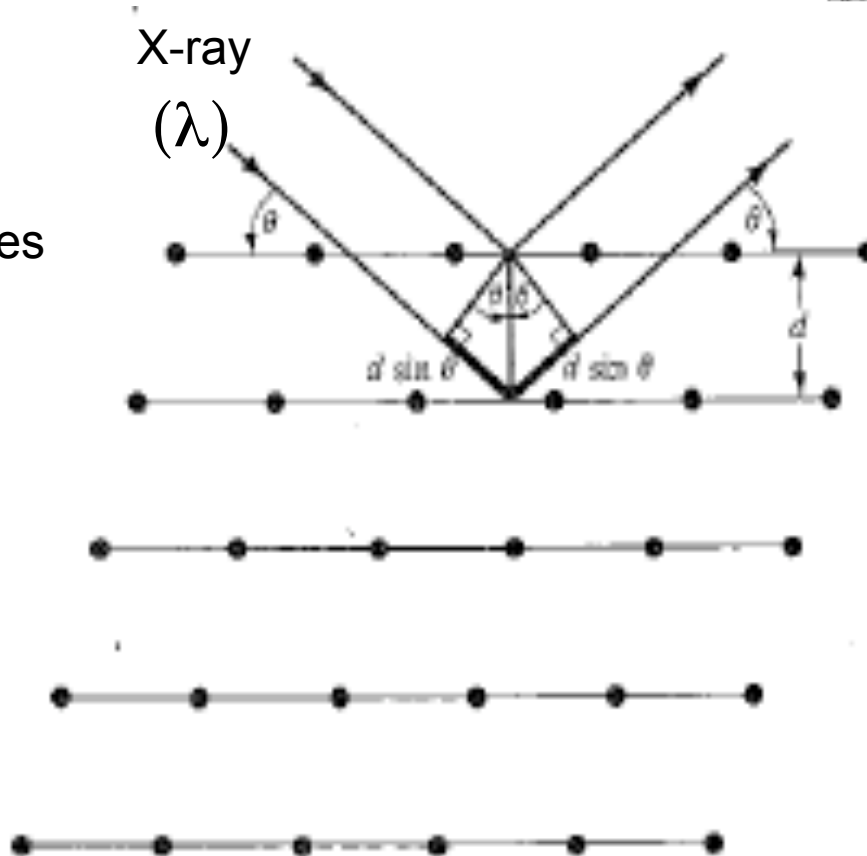
$$2d \sin \theta = n\lambda$$

(n = integer)



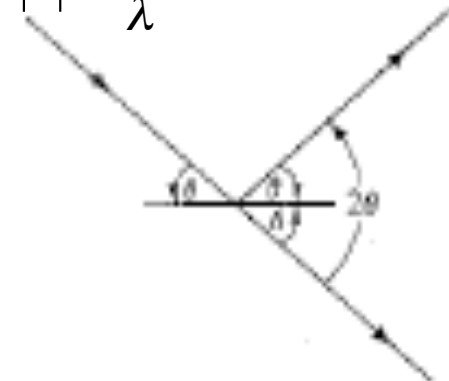
X-ray
(λ)

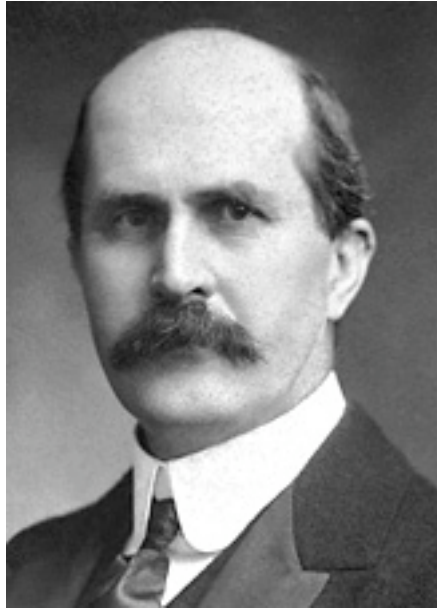
Lattice planes



Wave vector

$$|\mathbf{k}| = \frac{2\pi}{\lambda}$$





Sir W. Henry Bragg



W. Lawrence Bragg

1915:

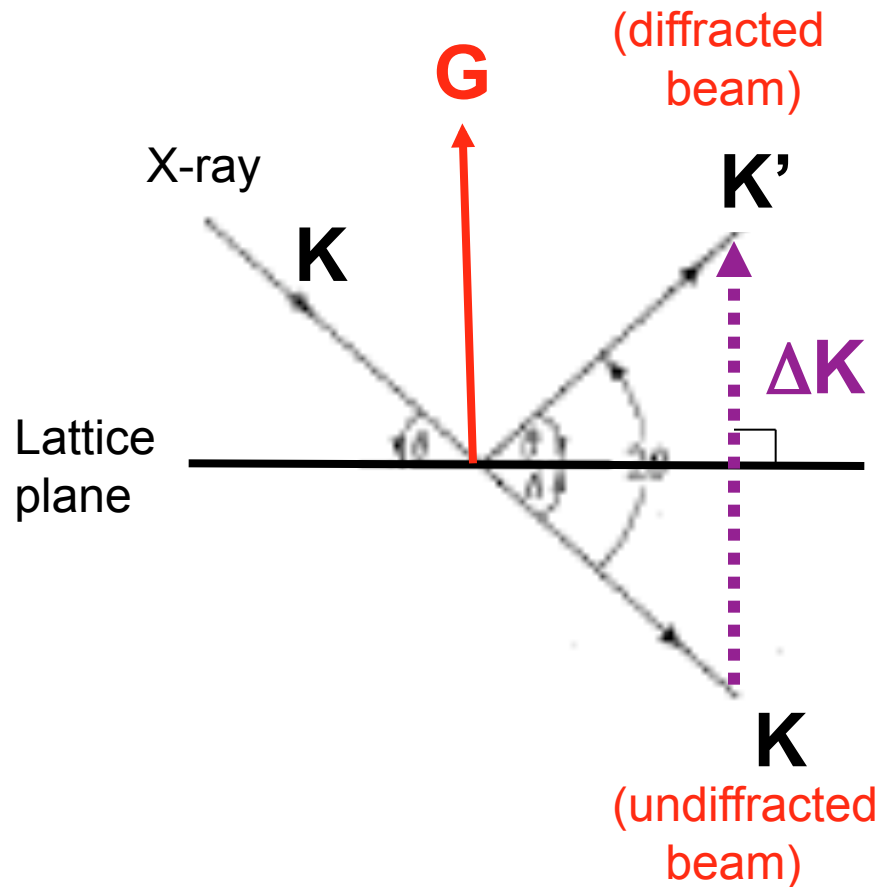
The Nobel Prize in Physics was awarded jointly to Sir William Henry Bragg and William Lawrence Bragg

"for their services in the analysis of crystal structure by means of X-rays"

Bragg Law:

$$2d \sin \theta = n\lambda$$

(n = integer)



Diffraction conditions:

The set of reciprocal lattice vectors \mathbf{G} determines the possible x-ray reflection

$$\Delta \mathbf{K} = \mathbf{K}' - \mathbf{K} = \mathbf{G}$$

$$\mathbf{K} + \mathbf{G} = \mathbf{K}'$$

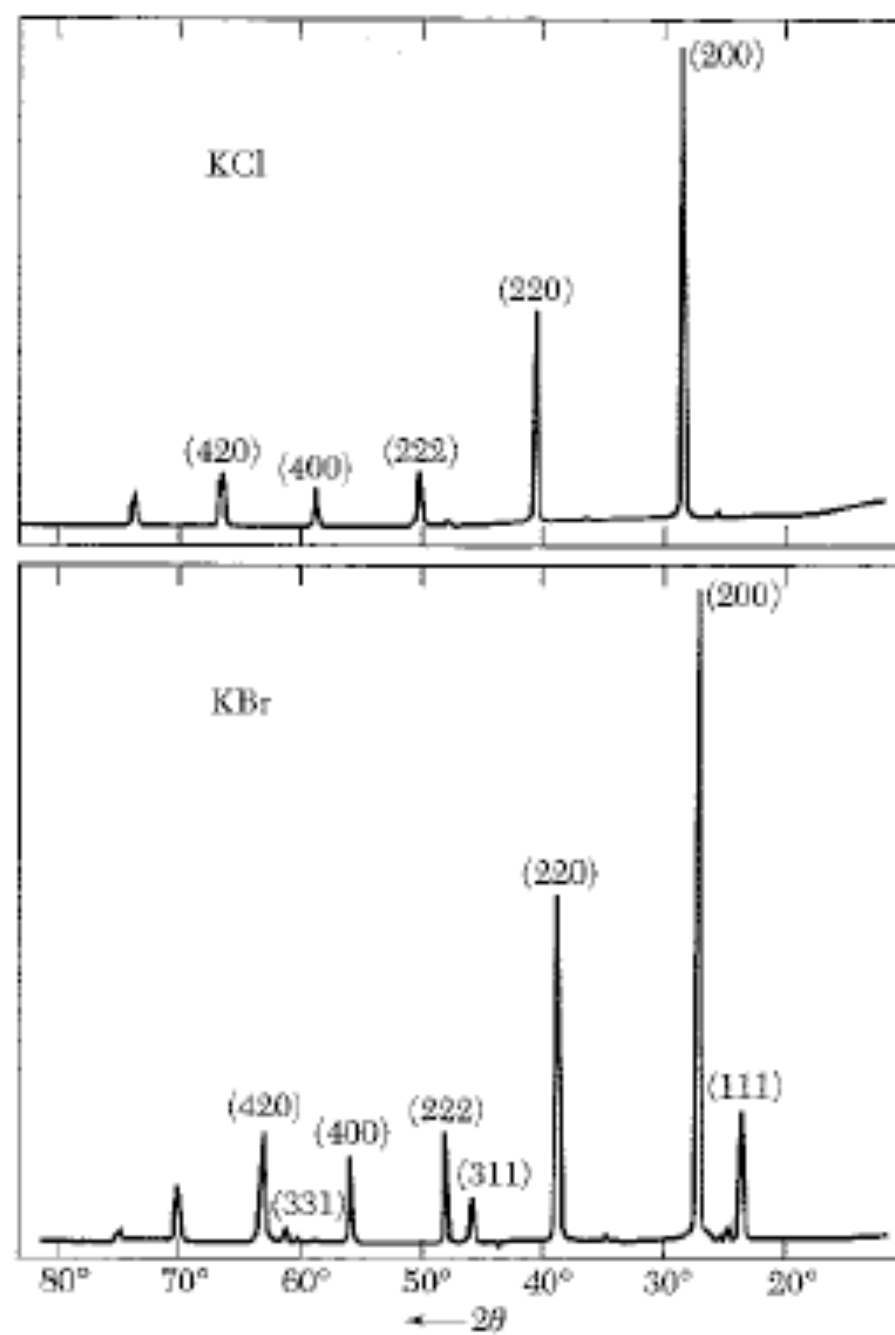
Elastic scattering: $K = K'$

$$(\mathbf{K} + \mathbf{G})^2 = K^2$$

$$2\mathbf{K} \cdot \mathbf{G} = G^2$$

$$\mathbf{G} = h\mathbf{b}_1 + k\mathbf{b}_2 + l\mathbf{b}_3$$

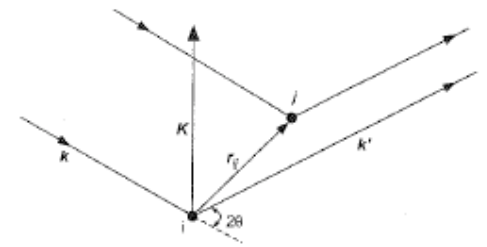
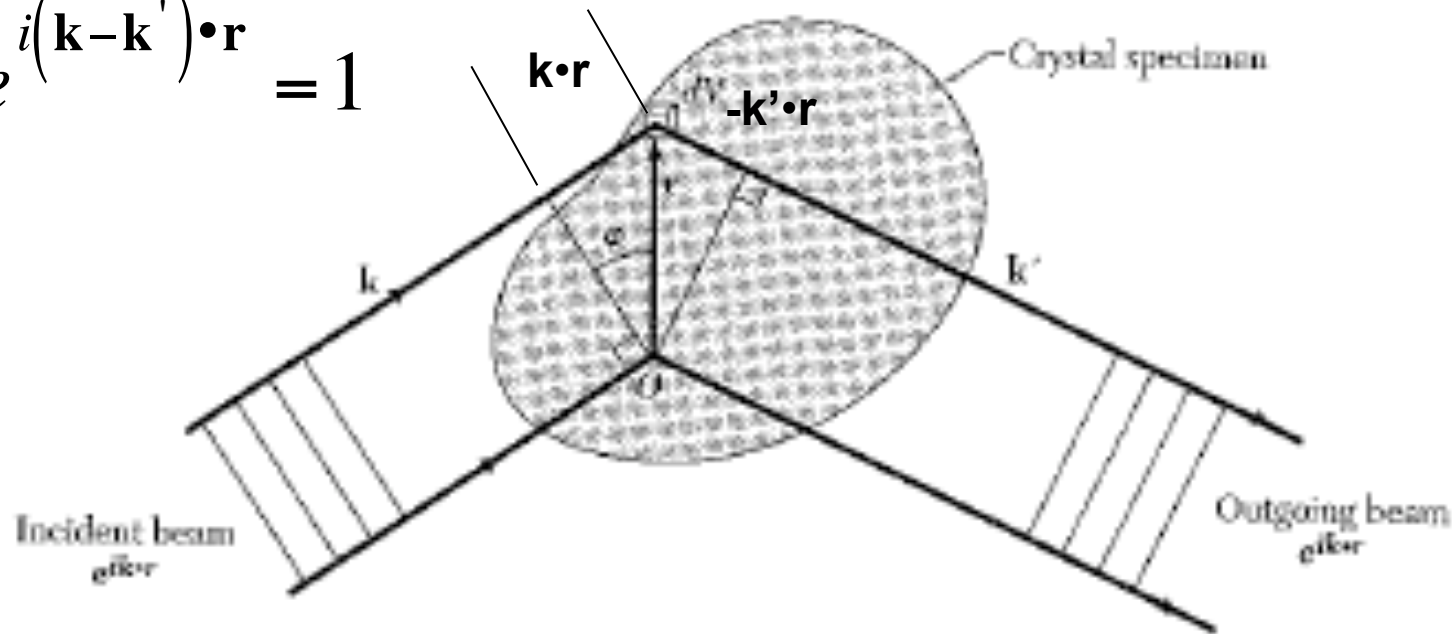
If cubic $d^{-1} \propto \sqrt{h^2 + k^2 + l^2}$



Von Laue Approach

Condition: $(\mathbf{K}-\mathbf{K}') \cdot \mathbf{r} = 2\pi n$ ($n = \text{integer}$)

$$e^{i(\mathbf{k}-\mathbf{k}') \cdot \mathbf{r}} = 1$$

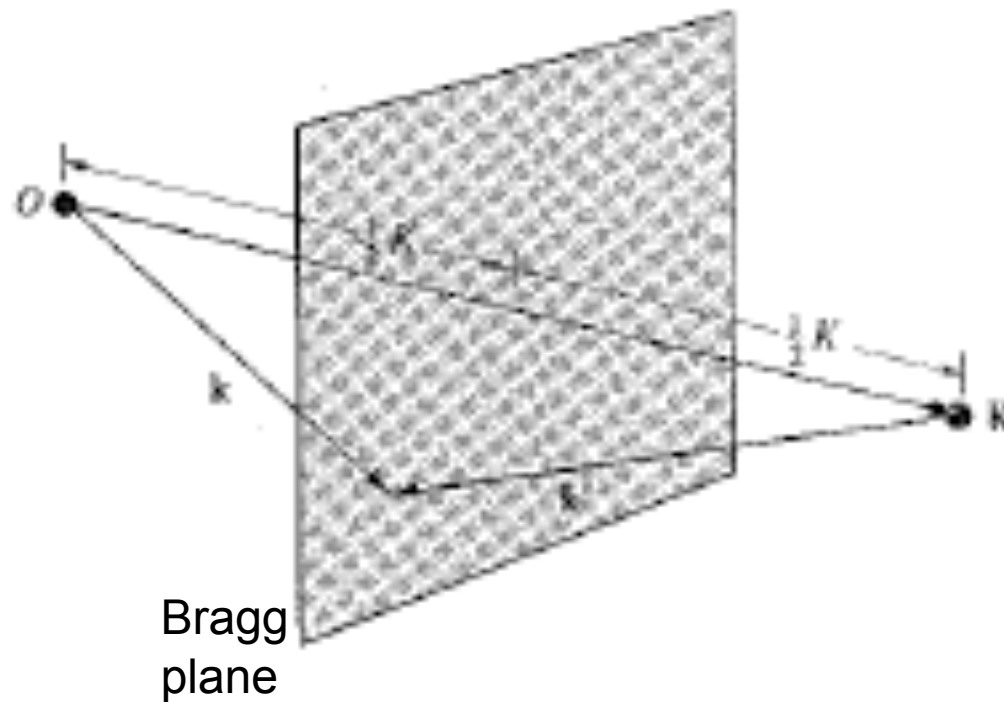


Constructive interference will occur if $\Delta\mathbf{K} = \mathbf{K}' - \mathbf{K}$ is a vector of the reciprocal lattice.

In the case of elastic scattering: $K = K'$

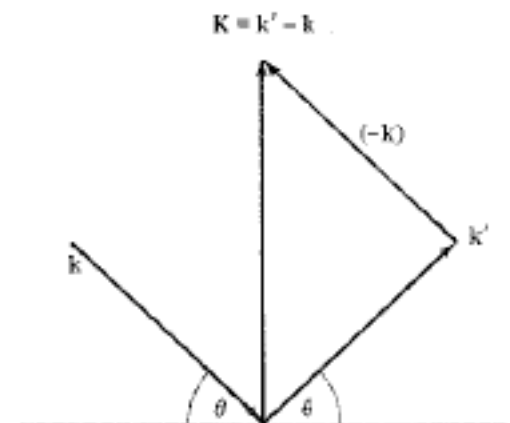
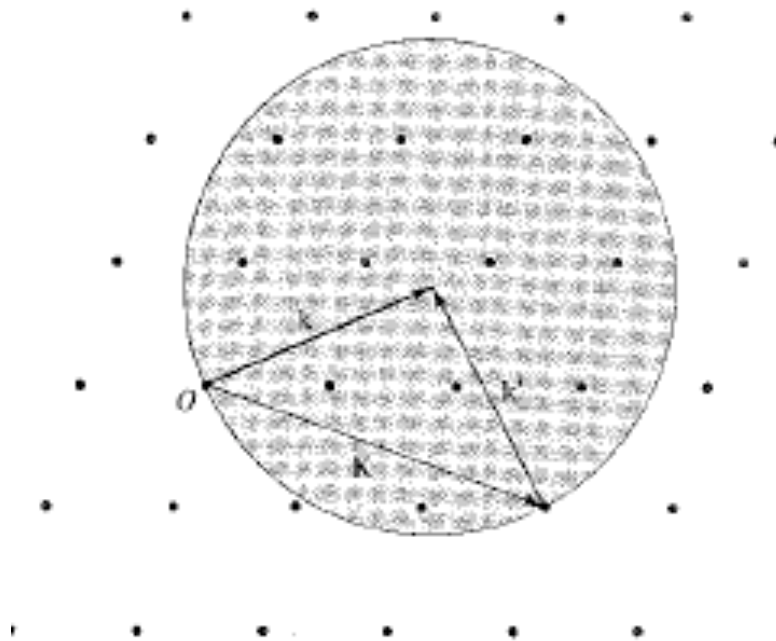
$$k = |\mathbf{k} - \Delta \mathbf{K}|$$

$$2\mathbf{k} \cdot \Delta \mathbf{K} = K$$



$\Delta \mathbf{K}$ is the reciprocal lattice vector: $K = n(2\pi/d) = 2k \sin \Theta$

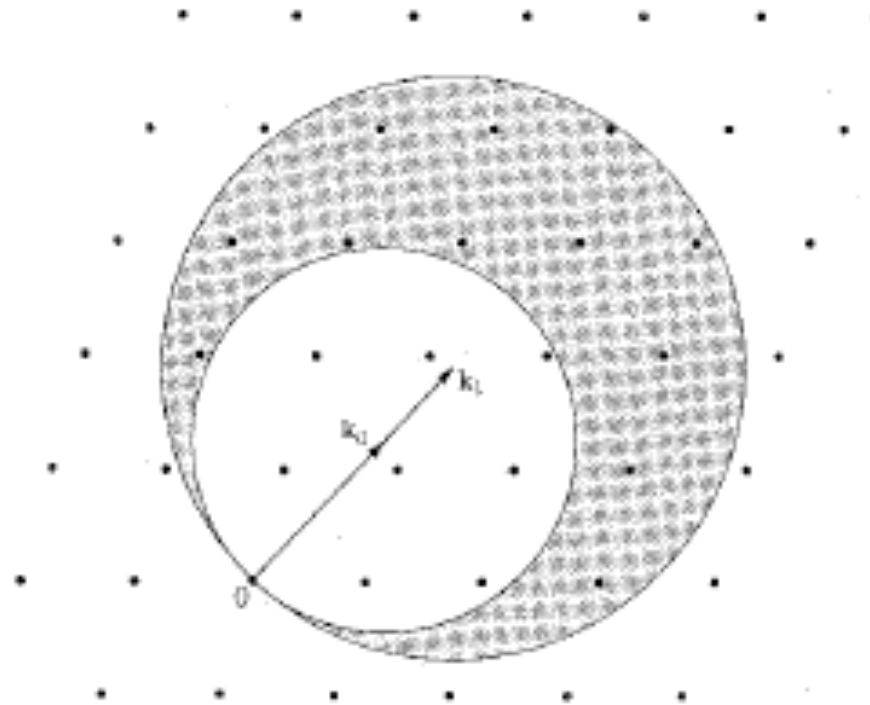
The Ewald Construction



The Laue Method

sample position and incident beam (direction) - unchanged

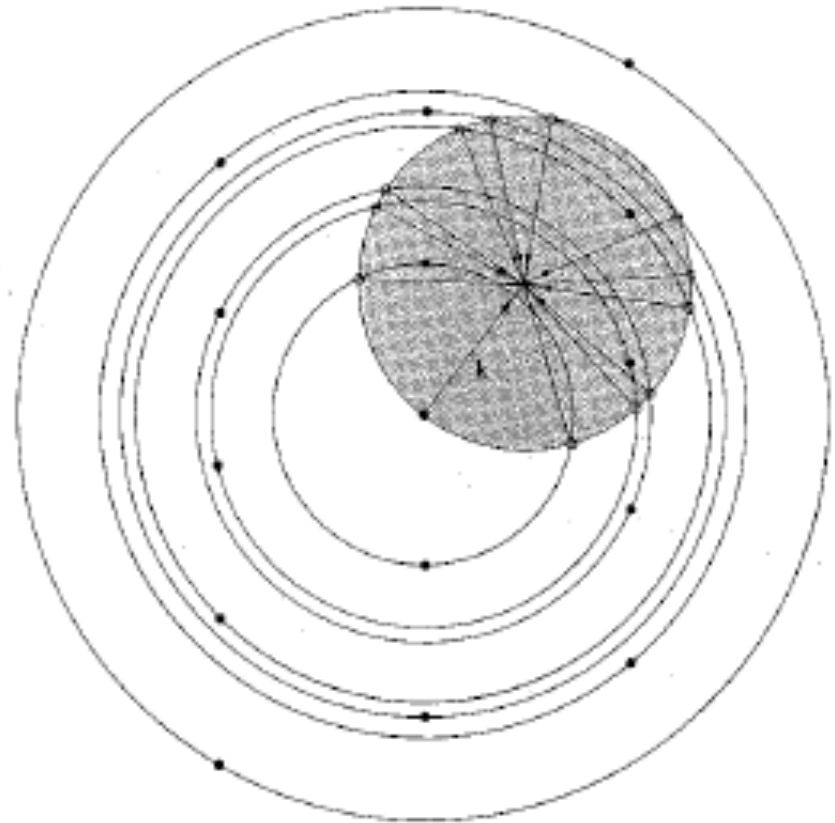
wave length - from λ_1 to λ_0



The Rotating - Crystal Method

Monochromatic beam (same λ)

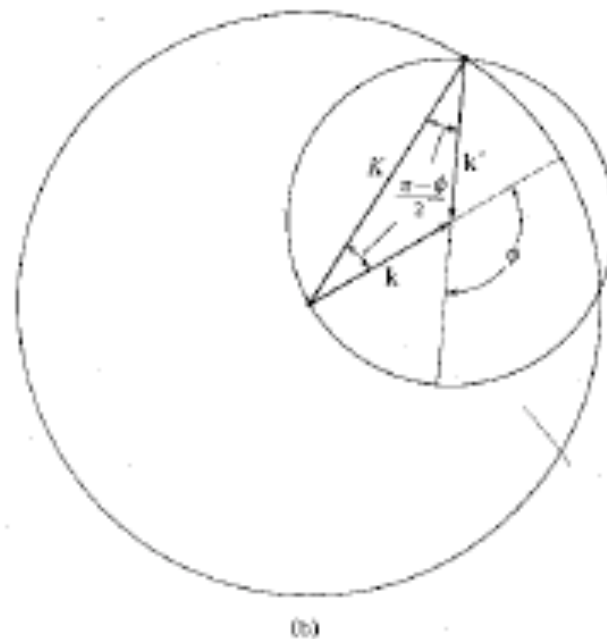
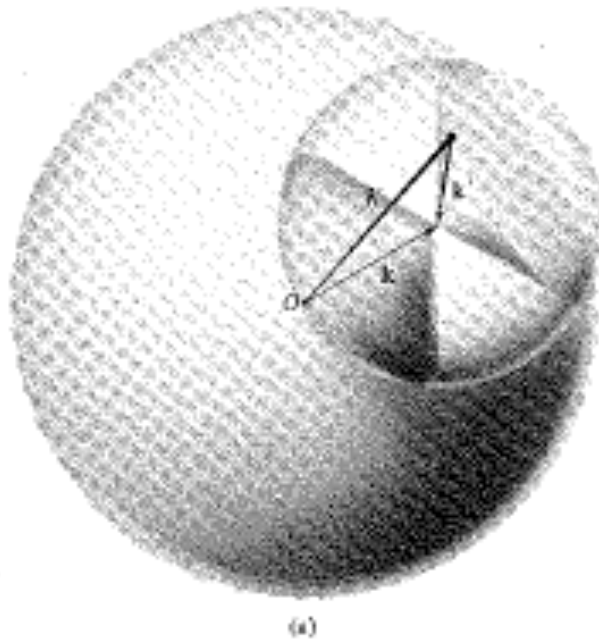
Rotate sample: change angle Θ



The Debye -Scherrer Method -- for Powder X-ray Diffraction

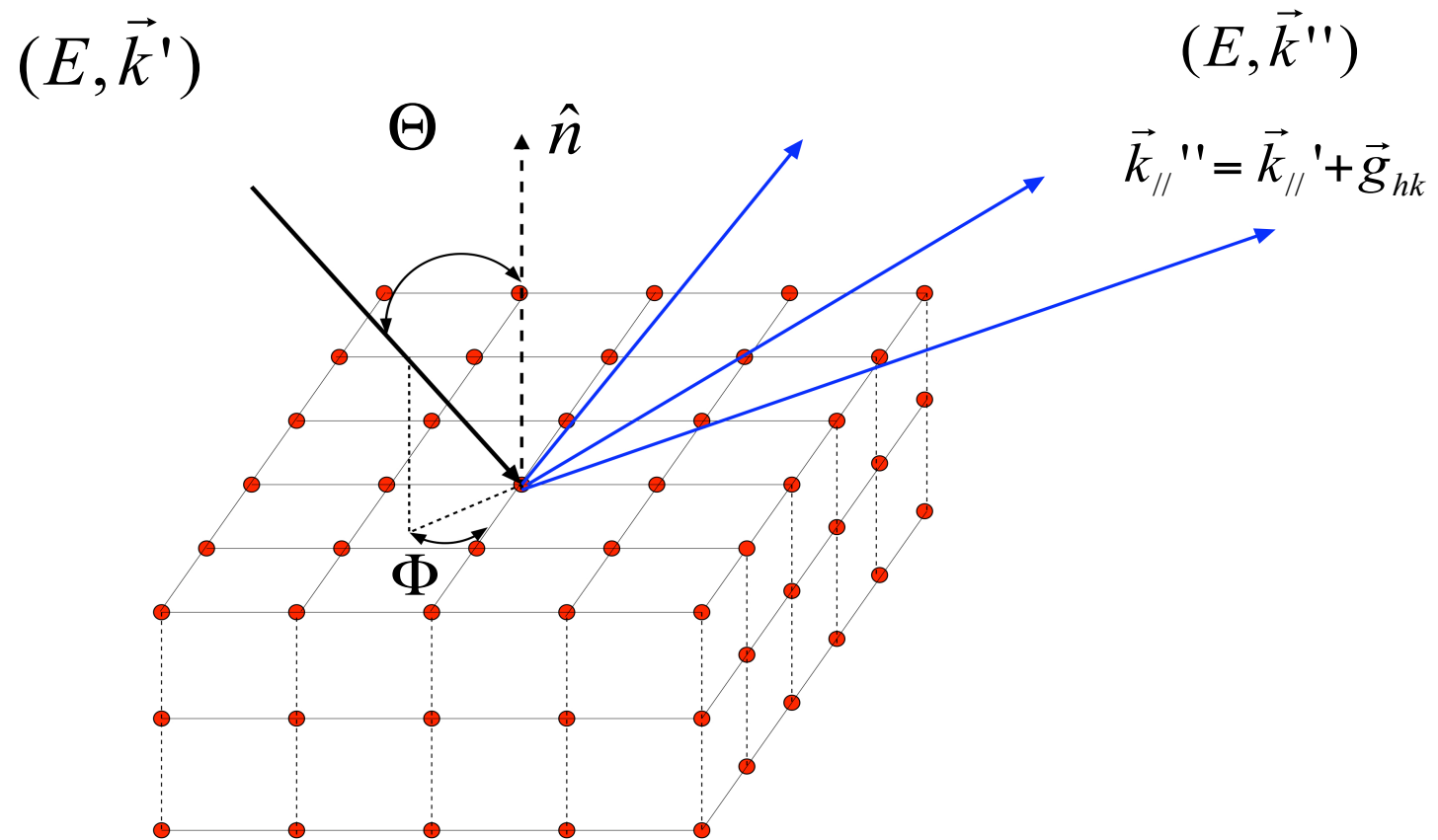
Monochromatic beam (same λ)

change angle Θ

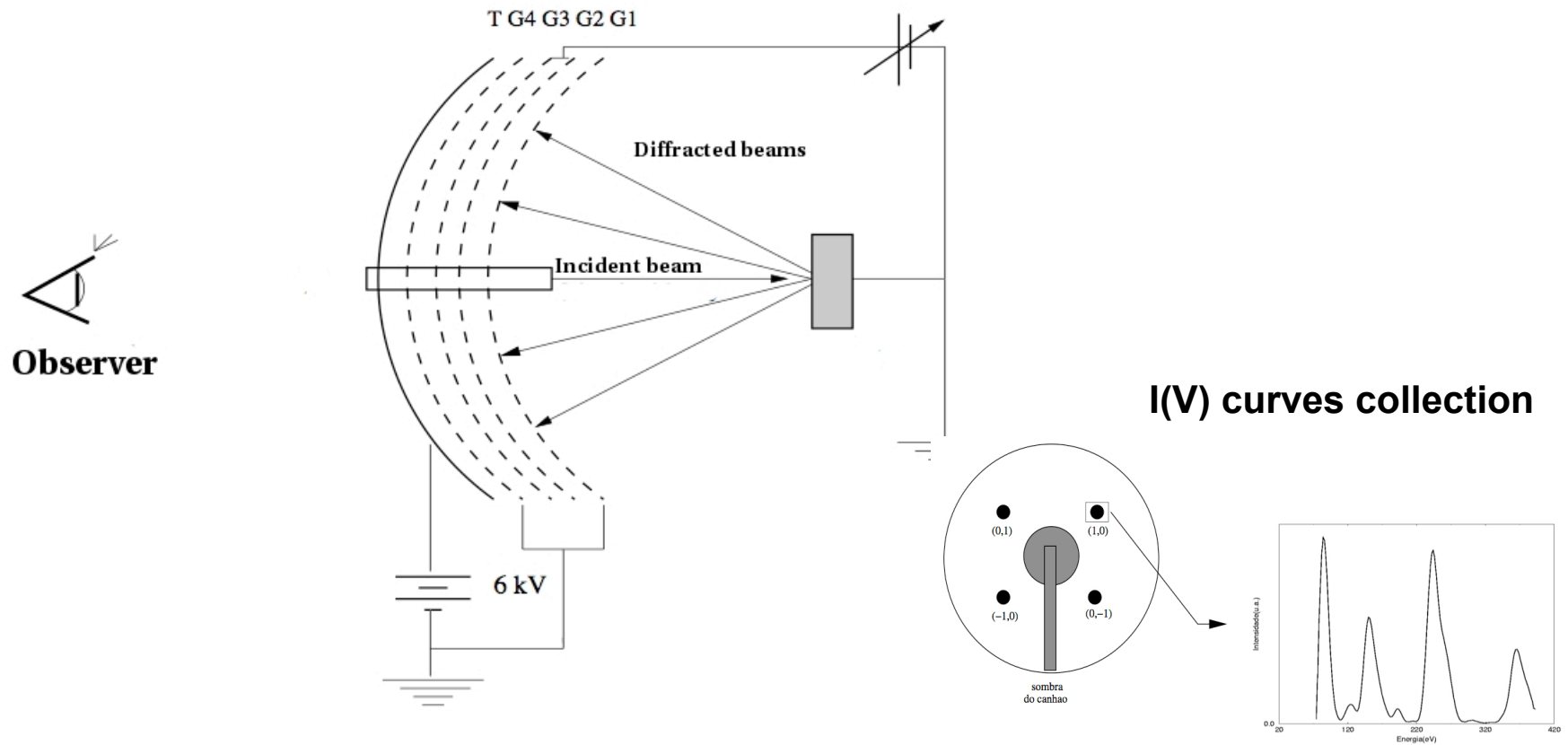


2D Structure Determination

Low-Energy Electron Diffraction

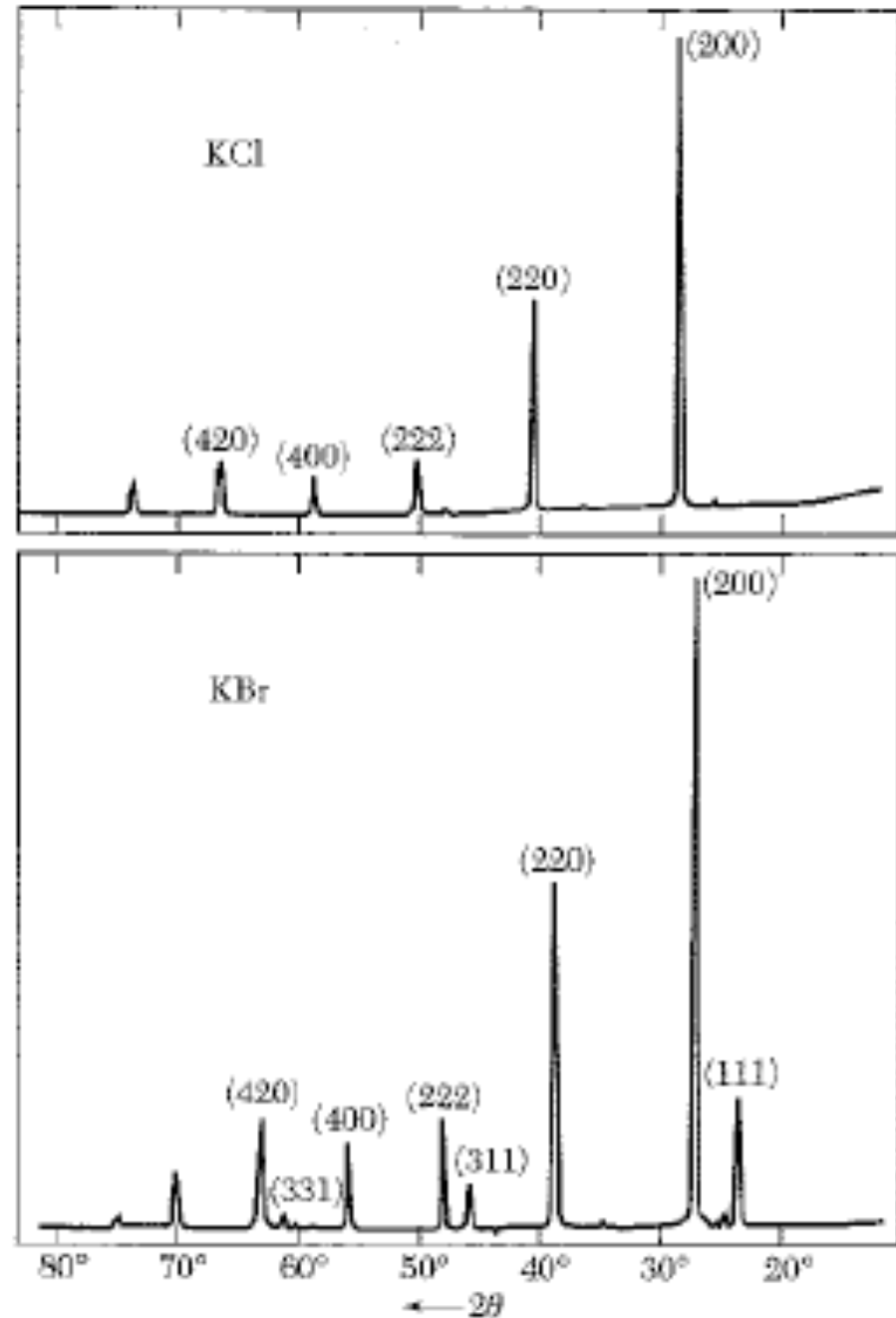


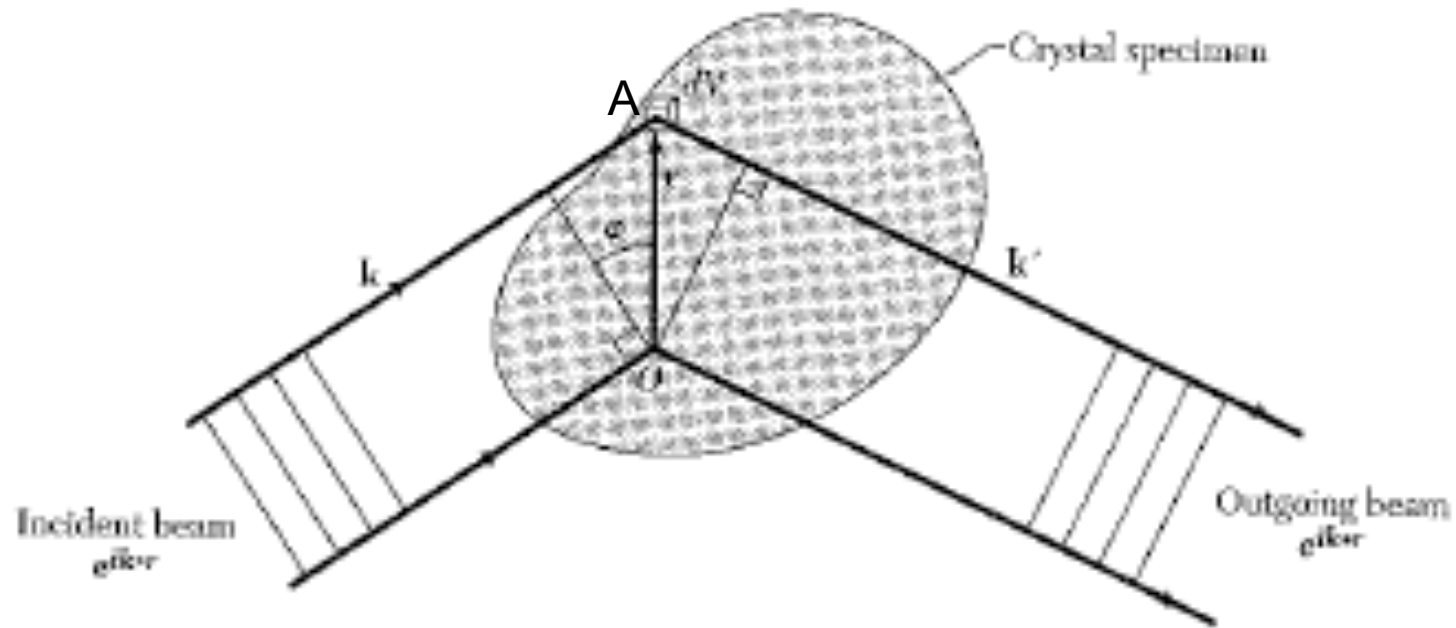
Retarding Field Analyzer



Are all Bragg peaks
revealed in the x-ray
diffraction pattern?

Why?





phase factor: $e^{i(\mathbf{k}-\mathbf{k}')\cdot\mathbf{r}}$

Scattering amplitude:
$$F = \int dV n(\mathbf{r}) e^{i(\mathbf{k}-\mathbf{k}')\cdot\mathbf{r}}$$

$$= NS_K$$

Structural factor:
$$S_K = \sum_{j=1}^n e^{i\Delta \mathbf{K} \cdot \mathbf{d}_j}$$

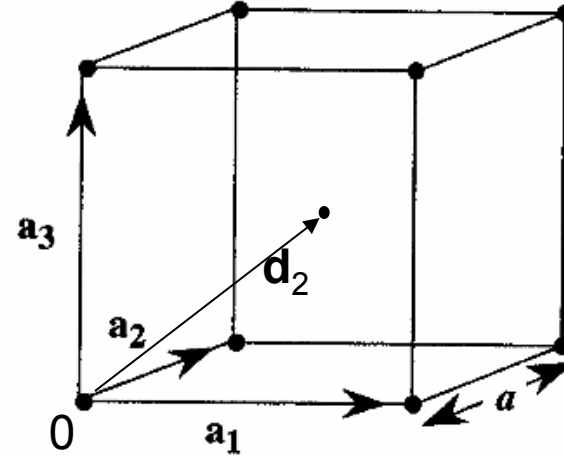
$\Delta \mathbf{K} = \mathbf{k} - \mathbf{k}'$ - vector of reciprocal lattice

\mathbf{d}_j - j^{th} basis point in Bravais lattice

Bragg peak intensity $\propto |S_K|^2$

Can predict when the peak vanishes

Body-centered cubic
Bravais lattice



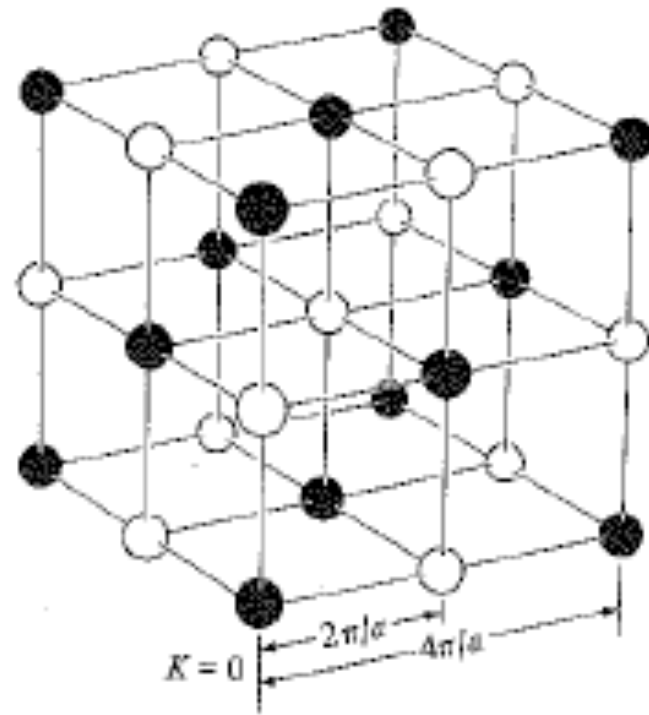
$$\mathbf{a}_1 = a\mathbf{x}, \quad \mathbf{a}_2 = a\mathbf{y}, \quad \mathbf{a}_3 = a\mathbf{z}$$

$$\mathbf{b}_1 = (2\pi/a)\mathbf{x}, \quad \mathbf{b}_2 = (2\pi/a)\mathbf{y}, \quad \mathbf{b}_3 = (2\pi/a)\mathbf{z}$$

Two basis points: $\mathbf{d}_1 = 0$, $\mathbf{d}_2 = (a/2)(\mathbf{x} + \mathbf{y} + \mathbf{z})$

$$\Delta\mathbf{K} = (2\pi/a)(n\mathbf{x} + m\mathbf{y} + l\mathbf{z})$$

$$S_K = \sum_{j=1}^n e^{i\Delta\mathbf{K} \cdot \mathbf{d}_j} = 1 + (-1)^{n+m+l} = \begin{cases} 2, & n+m+l = \text{even} \\ 0, & n+m+l = \text{odd} \end{cases}$$



$$S_K = \sum_{j=1}^n e^{i\Delta \mathbf{K} \cdot \mathbf{d}_j} = 1 + (-1)^{n+m+l} = \begin{cases} 2, & n+m+l = \text{even} \\ 0, & n+m+l = \text{odd} \end{cases}$$

$$\mathbf{a}_1 = (a/2)(\mathbf{y} + \mathbf{z}),$$

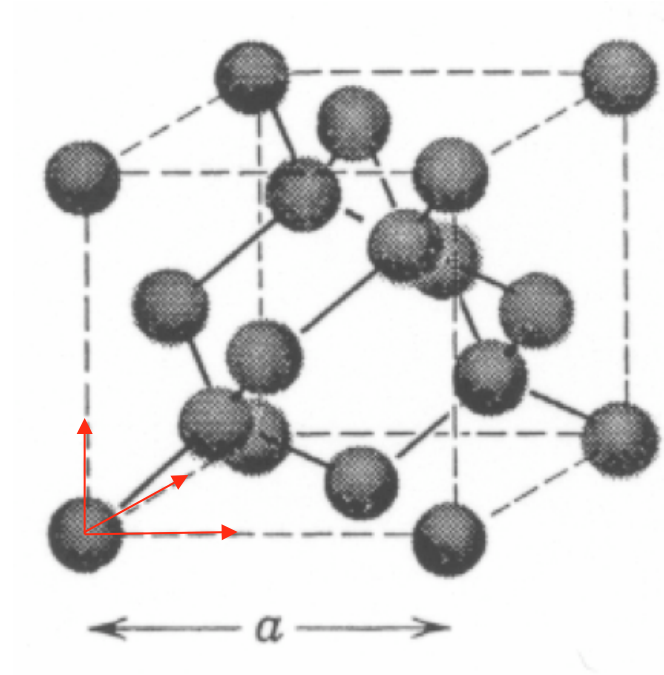
$$\mathbf{a}_2 = (a/2)(\mathbf{z} + \mathbf{x}),$$

$$\mathbf{a}_3 = (a/2)(\mathbf{x} + \mathbf{y})$$

$$\mathbf{b}_1 = (2\pi/a)(\mathbf{y} + \mathbf{z} - \mathbf{x})$$

$$\mathbf{b}_2 = (2\pi/a)(\mathbf{z} + \mathbf{x} - \mathbf{y})$$

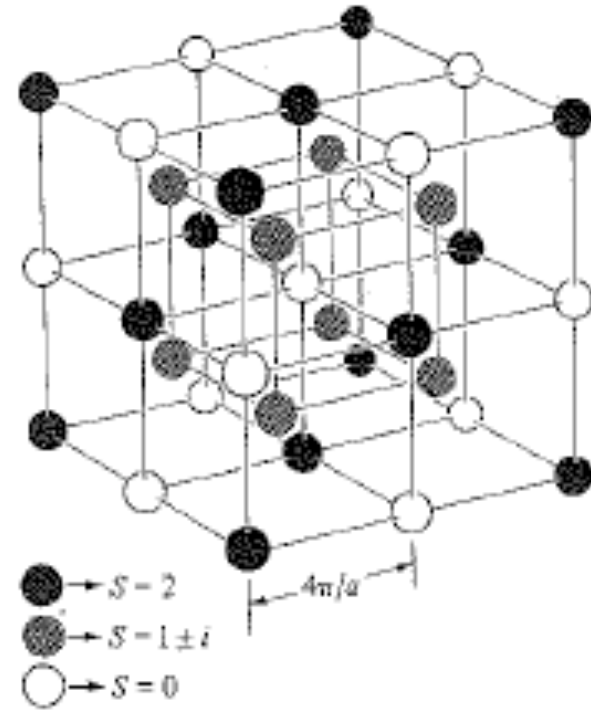
$$\mathbf{b}_3 = (2\pi/a)(\mathbf{x} + \mathbf{y} - \mathbf{z})$$



Two basis points: $\mathbf{d}_1 = 0$, $\mathbf{d}_2 = (a/4)(\mathbf{x} + \mathbf{y} + \mathbf{z})$

$$\Delta \mathbf{K} = (4\pi/a)(n\mathbf{x} + m\mathbf{y} + l\mathbf{z})$$

$$S_K = \sum_{j=1}^n e^{i\Delta \mathbf{K} \cdot \mathbf{d}_j} = 1 + e^{\left[\frac{1}{2}i\pi(n+m+l)\right]} = \begin{cases} 2, & (n+m+l)/2 = \text{even} \\ 1 \pm i, & n+m+l = \text{odd} \\ 0, & (n+m+l)/2 = \text{odd} \end{cases}$$



$$S_K = \sum_{j=1}^n e^{i\Delta \mathbf{K} \cdot \mathbf{d}_j} = 1 + e^{\left[\frac{1}{2}i\pi(n+m+l)\right]} = \begin{cases} 2, & (n+m+l)/2 = \text{even} \\ 1 \pm i, & n+m+l = \text{odd} \\ 0, & (n+m+l)/2 = \text{odd} \end{cases}$$

<i>Symmetry element</i>	<i>Reflection affected</i>	<i>Systematic-absence condition</i>
<i>Centred cells</i>		
Body-centred, I	hkl	$h + k + l = 2n + 1$
Face-centred, F	hkl	$h + k, h + l, k + l = 2n + 1$
Side-centred, C	hkl	$h + k = 2n + 1$
<i>Screw axis</i>		
2_1 along a	$h00$	$h = 2n + 1$
<i>Glide planes $\perp b$</i>		
Translation ($a/2$) (a -glide)	$h0l$	$h = 2n + 1$
Translation ($a/2 + c/2$) (n -glide)	$h0l$	$h + l = 2n + 1$
Translation ($a/4 + c/4$) (d -glide)	$h0l$	$h + l = 4n + 1, 2, 3$

In case of powder sample:

$$S_K = \sum_{j=1}^n f_j(\Delta \mathbf{K}) e^{i\Delta \mathbf{K} \cdot \mathbf{d}_j}$$

Atomic form factor

$$|S_K| \neq 0$$

Homework today (due on Sept. 9, 2010)

1. Problem 1 in page 93 (Ashcroft/Mermin)
2. Hexagonal space lattice: The primitive translation vectors of the hexagonal space lattice may be taken as: $\mathbf{a}_1 = (3^{1/2}a/2)\mathbf{x} + (a/2)\mathbf{y}$; $\mathbf{a}_2 = -(3^{1/2}a/2)\mathbf{x} + (a/2)\mathbf{y}$; $\mathbf{a}_3 = c\mathbf{z}$.
 - (a) Show that the volume of the primitive cell is $(3^{1/2}/2)a^2c$;
 - (b) Show that the primitive translations of the reciprocal lattice are
 $\mathbf{b}_1 = (3^{1/2}2\pi/a)\mathbf{x} + (2\pi/a)\mathbf{y}$; $\mathbf{b}_2 = -(3^{1/2}2\pi/a)\mathbf{x} + (2\pi/a)\mathbf{y}$; $\mathbf{b}_3 = (2\pi/c)\mathbf{z}$.
so that the lattice is its own reciprocal, but with a rotation of axes.